

WANL-PR(Q)-012 NASA-CR-72094

# DEVELOPMENT OF DISPERSION STRENGTHENED TANTALUM BASE ALLOY

Eleventh Quarterly Report

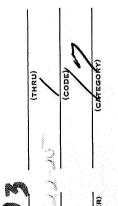
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R. W. Buckman and R. C. Goodspeed

prepared for
National Aeronautics and Space Administration
Lewis Research Center
Space Power Systems Division
Under Contract (NAS 3-2542)



ASTRONUCLEAR LABORATORY
WESTINGHOUSE ELECTRIC CORPORATION





FACILITY FORM 602



WANL-PR-(Q)-012 9 NASA-CR-72094

# DEVELOPMENT OF DISPERSION STRENGTHENED TANTALUM BASE ALLOY

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by

R. W. Buckman, Jr. and R. C. Goodspeed

Covering the Period

May 20, 1966 to August 20, 1966

Prepared For

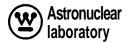
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#### **ABSTRACT**

Development of dispersion strengthened tantalum base alloys for use in advanced space power systems continued with the composition selection and melting of scale-up compositions Tα-8W-1Re-1Hf (ASTAR-811) and Tα-7W-1Re-1Hf-0.012C-0.012N (ASTAR-811CN) as 4 inch diameter ingots. The creep behavior of Tα-8W-1Re-0.7Hf-0.025C (ASTAR-811C) was affected by final annealing temperature with time to 1% strain at 2400°F and 15,000 psi increasing from 262 to 494 hours as the 1 hour pretest annealing temperature was increased from 3000°F to 3630°F. The time to 1% strain for TIG welds tested in the as-welded condition ut 2400°F and 15,000 psi was 171 hours.



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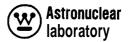


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#### I. INTRODUCTION

This, the eleventh quarterly progress report on the NASA-sponsored program "Development of a Dispersion Strengthened Tantalum Base Alloy", describes the work accomplished during the period May 20 to August 20, 1966. The work was performed under Contract NAS 3-2542.

The primary objective of the current phase of this program is the double vacuum arc melting of three compositions in the form **at** 60-pound, 4-inch diameter ingots. These compositions are to be selected for potential sheet and tubing application on the basis of their weldability, creep resistance, and fabricability characteristics.

Prior to this quarterly period, several promising tantalum alloy compositions were developed! These alloys exhibited good creep resistance at  $1315^{\circ}C$  (2400°F) while maintaining adequate fabricability. From these alloys, a weldable composition containing a carbide dispersion, Ta-8W-0.7Hf-1Re-0.025C (ASTAR-811C) was selected and melted as the first 4-inch diameter ingot (Heat NASV-20). The bottom portion of this ingot was upset forged and processed to 0.04 inch sheet. Another section was side forged and processed to 0.04 inch sheet and detailed information on processing characteristics was obtained.

During this quarterly period, the evaluation of the Ta-8W-0.7Hf-1Re-0.025C (ASTAR-811C) composition was essentially completed with the exception of a few remaining creep tests. Composition Ta-8W-1Re-1Hf (ASTAR-811) a solid solution strengthened alloy, and Ta-7W-1Re-1Hf-0.012C-0.012N (ASTAR-811CN) a carbonitride strengthened composition were both melted as 4-inch diameter ingots and processing to sheet was initiated. This completed melting of the scale-up compositions.

The evaluation of the creep behavior of composition Ta-8W-1Re-0.7Hf-0.025C (ASTAR-811C), heat NASV-20, continued. The creep resistance of as-TIG welded material was significantly less than the base metal.

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#### II. PROGRAM STATUS

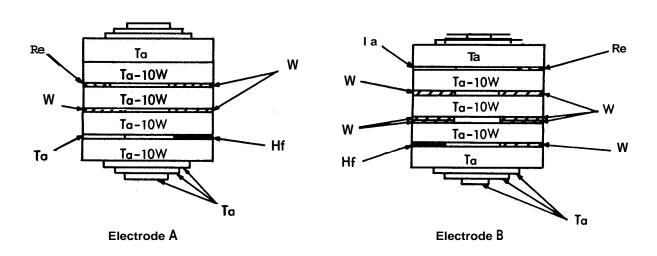
#### A. SCALE-UP

During this period, authorization was received from the cognizant NASA-project manager to proceed with the melting of the two remaining 4-inch diameter ingots. The compositions selected were:

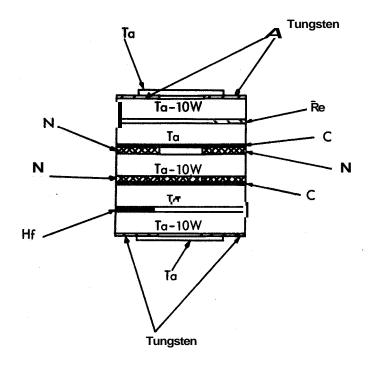
and are representative of solid solution strengthened and solid solution plus carbonitride strengthened alloys (3). The solid solution strengthened alloy composition has been designated ASTAR-811 and the 4-inch ingot was assigned heat number NASV-22. The solid solution plus carbonitride strengthened composition has been designated ASTAR-811CN and the 4-inch ingot was assigned heat number NASV-23. The carbide strengthened composition Ta-8W-1Re-0.7Hf-0.025C (ASTAR-811C) was melted as the first scale-up ingot and evaluation of this composition has been essentially completed.

1. Melting Two first melt electrodes, each weighing approximately 40 pounds were required for each heat. Sandwich type electrodes were fabricated from elemental metal strip. A cross-sectional view for each of the first melt electrodes is illustrated in Figure 1. The average cross-section composition of each electrode was at the nominal value for each constituent with the exception of the nitrogen addition to NASV-23. Loss of nitrogen during melting had been observed previously (4). Thus approximately 30% excess nitrogen was added to the first melt electrode to make up for the anticipated loss during melting.

The first melt electrodes were cast into a water cooled 2-1/2 inch diameter copper mold using AC power. A total of four first melt ingots were produced for each heat. The ends of the first melt ingots were faced, connected by studs, and welded to form the second melt electrode (6). Final melting was into a 4-inch diameter mold using DC power. First and second melt data for heats NASV-22 and 23 are listed in Table 1.



ASTAR-811
NASV-22 (Average Cross-Section Composition Ta-8W-1Re-1Hf)



ASTAR-811CN
NASV-23 (Average Cross-Section Composition To-8W-1Re-0. 95Hf-0. 012C-0. 016N)
(Carbon Added as Graphite Cloth and Nitrogen with Nitrided Tantalum Strip)

FIGURE 1 - Cross-Sectional Make-up of First Melt Electrodes



TABLE 1 - Melting Data for NASV-22 (ASTAR-811) and NAS-23 (ASTAR-811CN)

-			Mel	ting			
Melt No.	Electrode <b>I</b> dentification	Volts (AC)	Current (amps)	Melt Rate (bs/min)	Chamber Pressure (torr)	Dia. (in.)	Weight (lbs.)
ASTAR-811 CVAM-208 CVAM-209 CVAM-210 CVAM-211 CVAM-216	NASV-22A-1 NASV-22A-2 NASV-22B-1 NASV-22B-2 NASV-22	29 29 29 29 29 30(DC)	3,150 3,150 3,150 3,150 5,500(DC)	6.4 4.4 6.9 6.8 3.1 <b>(b)</b>	<5 x 10 <sup>-4</sup> a x 10 <sup>-4</sup> <5 x 10 <sup>-4</sup> <5 x 10 <sup>-4</sup> <5 x 10 <sup>-4</sup>	2.5 2.5 2.5 2.5 4.0	23.6 18.6 24. 1 17.6 69.7
ASTAR-811C CVAM-212 CVAM-213 CVAM-214 CVAM-215 CVAM-217	NASV-23B-1 NASV-23B-2 NASV-23A-1	29 29 29 29 29 30(DC)	3,150 3,150 3,150 3,150 5,500 (DC)	6.5 5.9 6.2 6.2	<5 x 10 <sup>-4</sup> <5 x 10 <sup>-4</sup> <5 x 10 <sup>-4</sup> <5 x 10 <sup>-4</sup> <5 x 10 <sup>-4</sup>	ノカー	23.4 16. 4 23.6 17. 2 55. 0(a)

<sup>(</sup>a) Conditioned ingot weight with hot top removed.

<sup>(</sup>b) Based on total melt duration which includes time for hot topping.



Samples were removed from the top and bottom of each ingot and analyzed for the metallic and interstitial additions. The chemical analysis results on the second melt ingots in Table 2 are indicative of the excellent compositional control obtained with the melting practice used.

TABLE 2 - Chemical Analysis Results for Heats NASV-22 and 23

Nomina   Composition	Ingot		Chem	ica l An	a lysis, we	ight perce	ent
(w/o)	Location	W	Hf	Re	С	Ν	O
Ta-8W-1Re-1Hf(NASV-22)	Top	7.8	0.95	1.04	0.0009	0.0010	0.0012
	Bottom	7.5	1.06	0.99	0.0016	0.0013	0.0020
Ta-7W-1Re- 1Hf-	Top	6.5	0.98	1.05	0.013	0.013	0.0018
0.012C-0.012N(NASV-23)	Bottom	6.4	1.01	1.01	0.011	0.011	0.0018

Each ingot was lathe conditioned and the hot-top and starting pads were removed.

A 1 inch slice was then cut from the bottom portion of the ingot. The ingots after conditioning are shown in Figure 2. The conditioned weight of NASV-22 was 61 pounds and NASV-23, 55 pounds, representing yields from the first melt electrode of 72% and 68% respectively.

2. <u>Ingot Microstructure</u> — A sample removed from the bottom of each ingot was mounted and examined metallographically. Photomicrographs of the as-cast microstructure are shown in Figures 3 and 4. The microstructure of the Ta-8W-1Hf-1Re (NASV-22) is essentially single phase with a minor amount of grain boundary precipitate observable. This precipitate is most probably HfO<sub>2</sub>. The microstructure of the carbonitride strengthened composition, Ta-7W-1Re-1Hf-0.012C-0.012N (NASV-23) contains a well defined dispersed second phase which is most likely the dimetal carbide (Ta<sub>2</sub>C). The identity of this precipitate is being confirmed. The as-cast hardness of NASV-22 was 231 DPH while that of NASV-23 was 300 DPH.



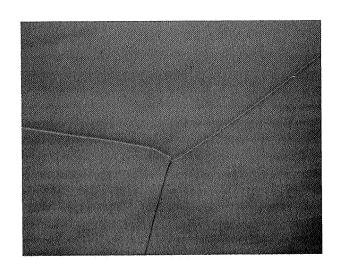


FIGURE 2 - Conditioned Four Inch Diameter Consumable Electrode
Double Vacuum Arc Melted Ta Alloy Ingots

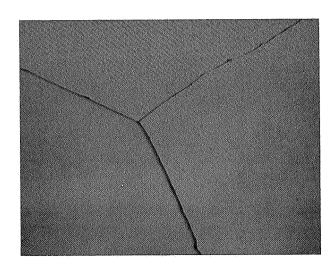
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a. 150X



b. 500X

FIGURE 3 - Microstructure of As-Cast Ta-8W-1Re-1Hf (ASTAR-811) Heat NASV-22 Etchant (25% HNO  $_3$  - 25% HF - 50% Glycerine) Oblique Lighting



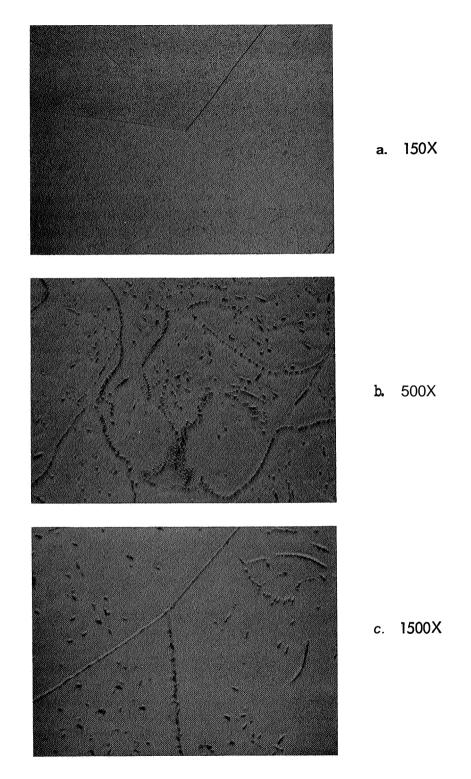


FIGURE 4 - Microstructure of As-Cast Ta-7W-1Re-1Hf-0.012C-0.012N (ASTAR-811CN) Heat NASV-23. Etchant (25%, HNO<sub>3</sub>-25% HF-50% Glycerine). Oblique Lighting



3. Primary Working — Four inch diameter x one inch thick billets, each weighing 6-1/2 pounds, from NASV-22 and 23 were coated with AI-12Si and then upset forged at 1400°C in a single blow on the Dynapak. The forging data are in Table 3. Both compositions exhibited excellent forgeability characteristics. There were no defects observed on either of the as-forged billets. The as-forged billets are being conditioned and will then be annealed at 1650°C (3000°F) for 1 hour prior to processing to sheet. The processing schedule of the NASV-22 and 23 will be the same as that used for ASTAR-811C (Heat NASV-20)(5).

#### 4. Mechanical Property Evaluation

a. ASTAR-811C (Heat NASV-20) — The creep behavior of heat NASV-20 of composition Ta-8W-1Re-0.7Hf-0.025C (ASTAR-811C) has been investigated over the temperature range of 2200-2800°F. The data accumulated thus far are listed in Table 4 and plotted in Figure 5 using the Larson-Miller parameter. Increasing the final annealing temperature results in an increase in creep resistance which has been shown to follow a functional relationship of the square root of grain diameter (3). The recrystallized grain diameter is also affected by the prior mechanical history. For sheet material annealed 1 hour at 3000°F, the recrystallized grain diameter increased from 0.017 mm for sheet reduced 83% before annealing to 0.033 mm for a prior reduction of 33% (3). TIG welding also caused a significant decrease in the 2400°F creep behavior. The time to elongate 1% at 2400°F and 15,000 psi for the TIG weld was 171 hours compared to 262 hours for the base material. The two tests listed in Table 4 are still in progress and will be continued until rupture so that the fracture behavior can be studied.

#### 111. FUTURE WORK

During the next quarterly period, we plan to accomplish the following:

- 1. Process the upset forged billets of NASV-22 and NASV-23 to 0.04 inch sheet and initiate mechanical property evaluation.
  - 2 Initiate response-to-heat treatment investigation of NASV-22 and NASV-23.
  - 3. Complete creep property evaluation of ASTAR-811C (Heat NASV-20).



TABLE 3 - Forging Results for Compositions Ta-8W-1Re-1Hf (NASV-22) and Ta-7W-1Re-1Hf-0.012C-0.012N (NASV-23)

F	orging	Billet	. :		As F	orged Billet	
Heat No.	Dia.	ensions Height (in.)	Hardness (DPH)	Forging Temperature (°C/°F)	Dia. Height (in.) (in.)	Hardness (DPH)	Upset (%)
NASV-22	4	1	231	1400/2550	5-1/4 0.44	292	56
NASV-23	4	1	300	1400/2550	5-1/4 0.44	355	56



TABLE 4 - Creep Results for ASTAR-811C, Ta-8W-1Re-0. 7Hf-0. 025C (Heat NASV-20)

	ŀ					T. O. T.		Hardness (DPH)	
	Annealina	Grain		Test		Elongate	Pre-Test	Post-Test	Test
T * erature	Temp.	Size	Stress	Duration	Elongation	]%		Head	Gage
Ŧ	( <sup>O</sup> F)	(mm)	(jsi)	(hours)	(%)	(hours)		Section	Length
2400	3000	0.033	12,500	531	0.93	220ها	255		 
2400	3000	0.033	15,000	554	2.53	262	249	cy cy	;
2400	3000	0.017	15,000	457 <sup>(c)</sup>	3.0	202	•		
2200	3000	0.017	18,000	1,017	0.1	1,017	255	218	217
2600	3000	0.017	8,000	165	0.88	188(0)	255	228	239
2600	3000	0.017	8,000	283	2. 62	159	255	235	238
2800	3000	0.017	4,000	260	ი ი	78	253	230	244
2400	3270	0.063	15,000	202	2.0	280	249	526	1
2400	3630	0.2	15,000	1,003,	2.1	474	<b>5</b> 60	ଷ	I I
2600	3630	0.2	8,000	719(c)	4. 17	296	1	I I	
2400	(9)	!	15,000	929	6.03	171	265	227	!
2600	<b>(</b> 9)	!	10,000	194	4.2	69.5	263	232	236

NOTE:
(a) Extrapolated value
(b) As-TIG welded, tested with weld bead in longitudinal direction
(c) Test in progress



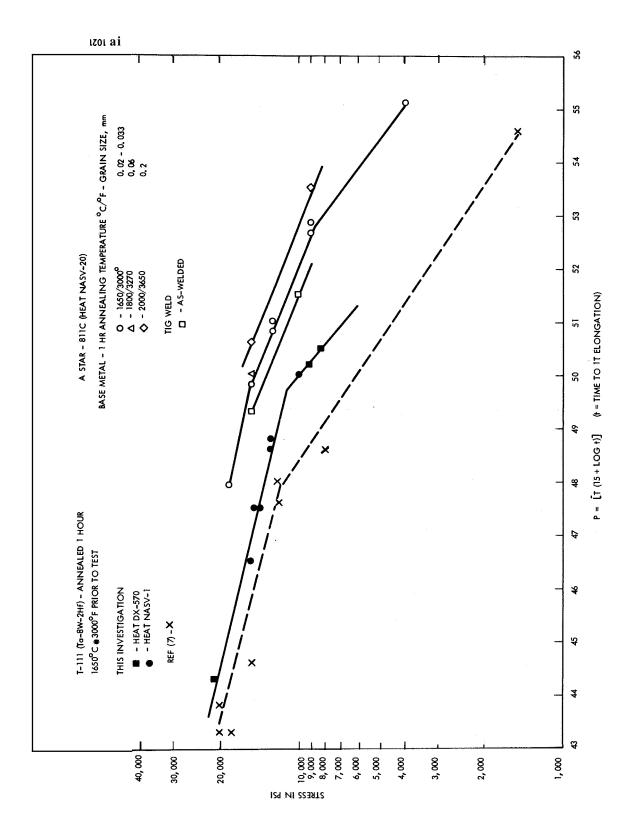


FIGURE 5 - Creep Behavior of Ta-8W-1Re-0. 7Hf-0. 025C (ASTAR-811C), Heat NASV-20



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